

Preparation and characterization of nanostructured thermoelectric materials

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$\text{Ca}_x\text{Co}_4\text{Sb}_{12}$ /substrate films , BiSbTe nano-powders

Bi/PbTe/substrate films

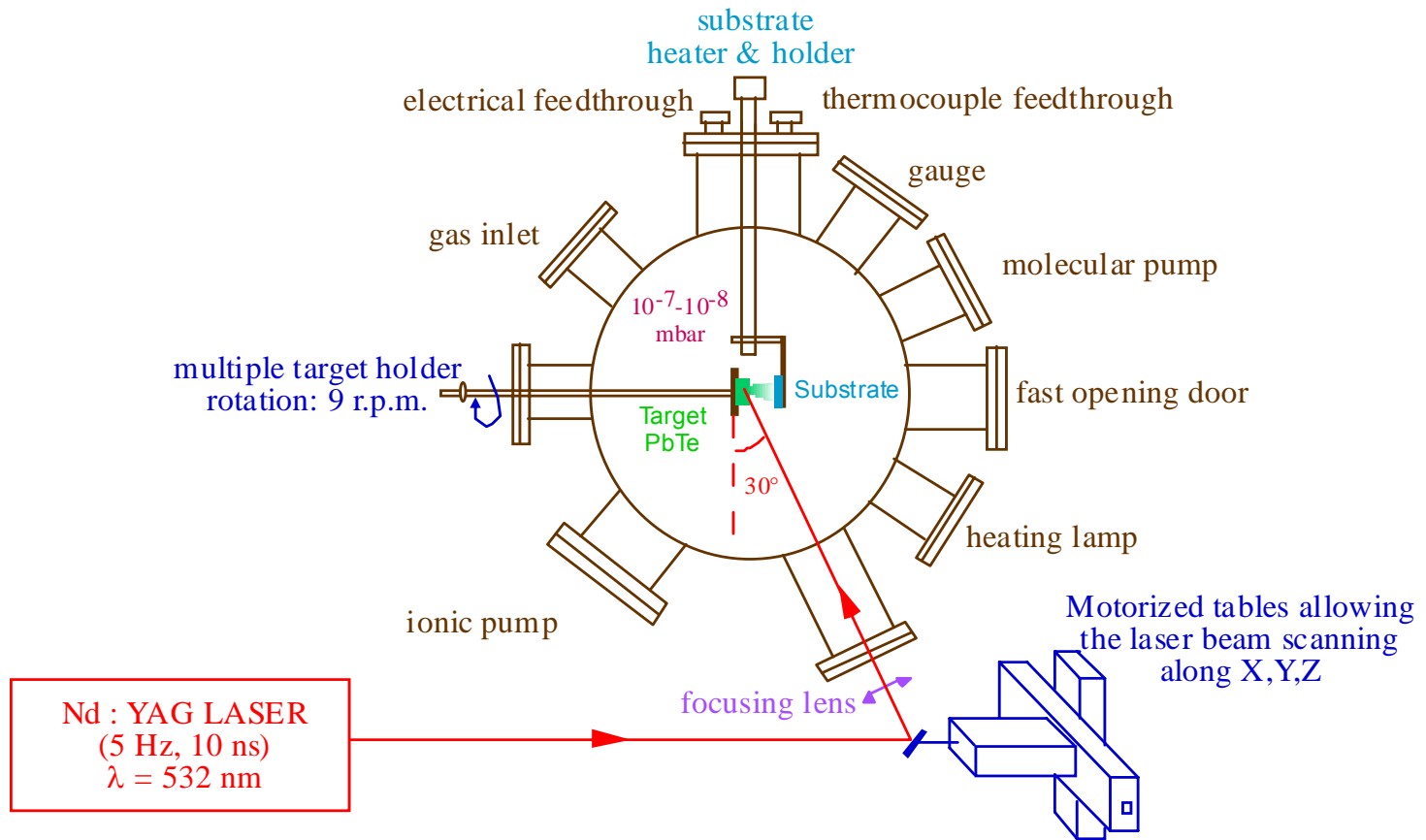
Film specifications

- high quality films
- well-known thickness
- high density
- large grains
- smooth surfaces
- sharp interfaces - no interdiffusion
- no oxygen contamination

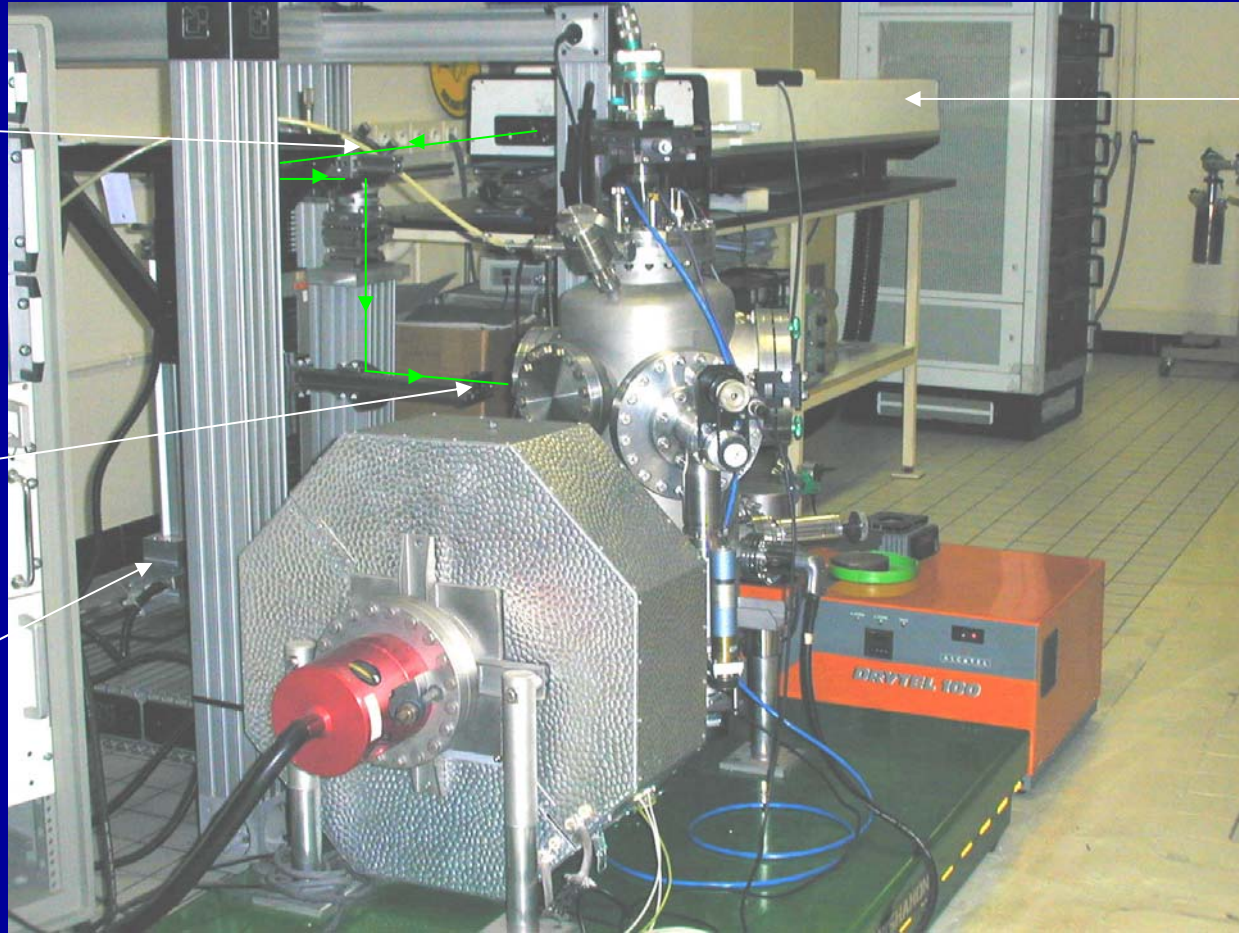
Pulsed laser deposition (PLD) for the synthesis of thermoelectric films

Stoichiometry of target restored
Lower deposition temperature

Experimental: pulsed laser deposition set-up



Experimental: pulsed laser deposition set-up



Laser beam

Nd:YAG laser

Focusing lens

x,y,z motorized tables

Adjustable parameters

Laser	Wavelength: 266, 532, 1064 nm (absorption) Frequency: 2, 5 or 10 Hz (overlapping) Output energy/focalisataion (fluence) Number of shots (thickness)
Chamber	Vacuum (static or dynamic) Gas (static or dynamic)
Target	Polycrystalline self-made ingots Diameter (12-19 mm) Rotation speed
Substrate	Nature (amorphous, oriented) Temperature (20-500°C) Target-substrate distance (2-5 cm)
Scanning	Rate

Preparation of skutterudite thin films

AIM: Synthesis of high quality *n*- and *p*-type skutterudites films for their use in thermoelectric micro-devices. Study of the influence of many deposition parameters to achieve a single phase film having the skutterudite structure.

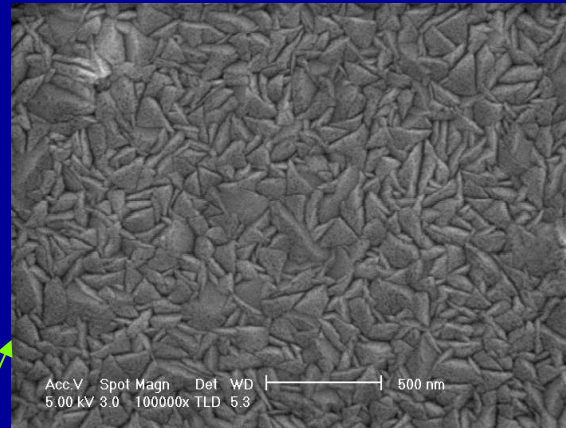
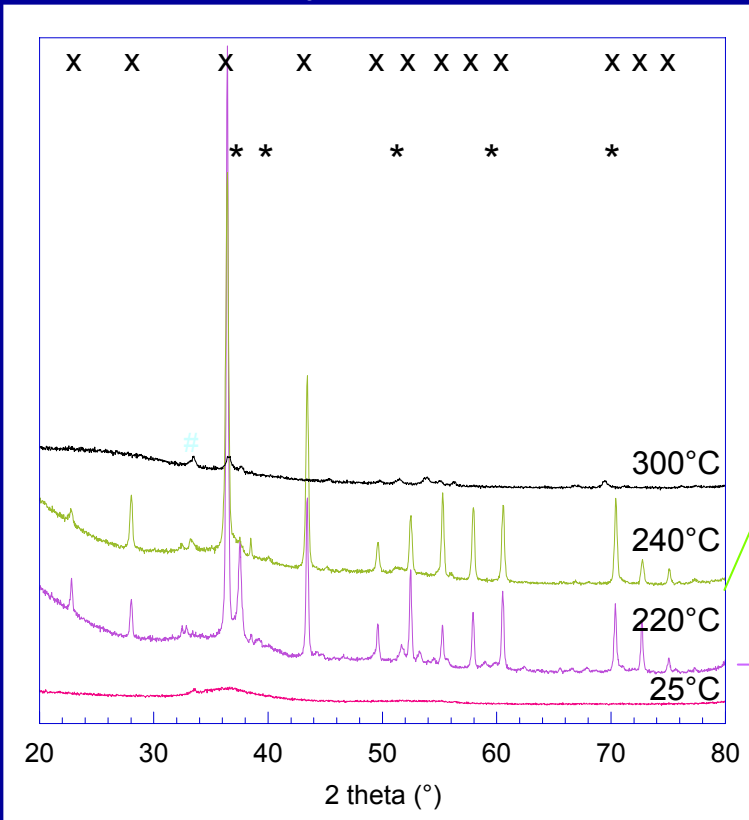
MATERIAL: $\text{Ca}_x\text{Co}_4\text{Sb}_{12}$ (*n*-type), $\text{Ce}_x\text{Fe}_{3.5}\text{Co}_{0.5}\text{Sb}_{12}$ (*p*-type)

PARAMETER STUDIED: wavelength: 266, 355 or 532 nm, density of energy: 2-5 J/cm², deposition temperature: 25-300°C,

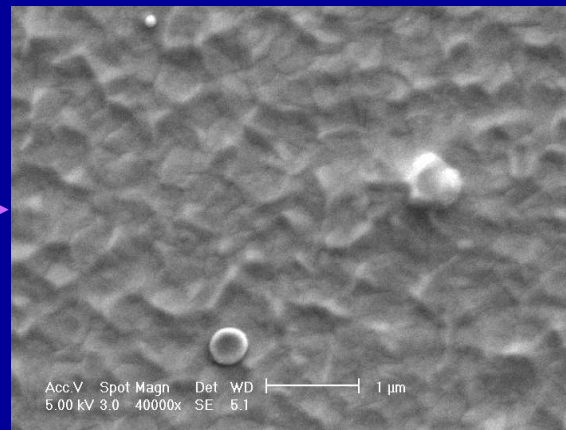
SUBSTRATE: SiO₂(20 nm)/Si(100), quartz, glass

Influence of the deposition temperature (532 nm)

X: CoSb_3 , *: CoSb_2 ?, #: Sb



Thickness: 520 nm
Grain: ~ 100 nm
RMS = 38.4 nm

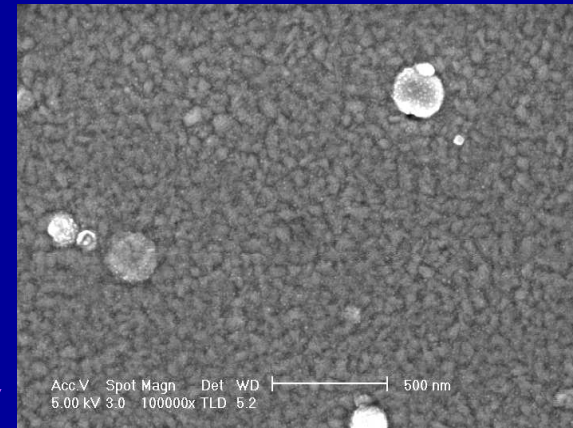
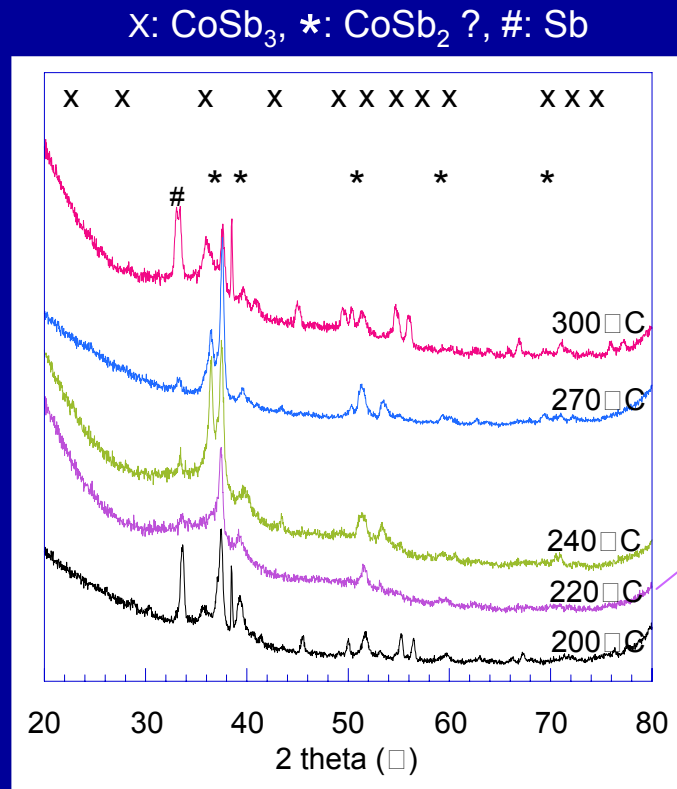


Thickness: 670 nm
Grain: > 500 nm
RMS = 16.1 nm

300°C: textured CoSb_3 + ?
240°C: CoSb_3 single phase
220°C: CoSb_3 + ?
25°C : amorphous

Decrease of film thickness and grain size
when $T_s \uparrow$

Influence of the deposition temperature (266 nm)

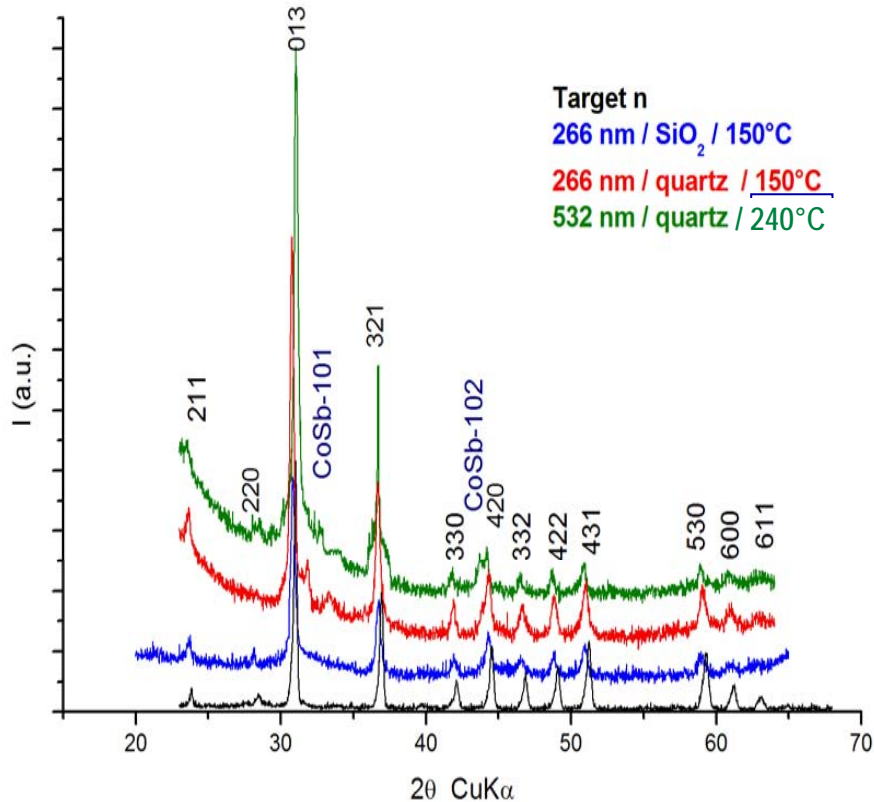


Thickness: 70 nm
Grain size: < 80 nm
RMS: 12.7 nm

Comparison with 532 nm:

- No achievement of the skutterudite phase, whatever T_s or fluence (\neq from the literature)
- No reduction of the droplet density
- Deposition rate about 10 times lower

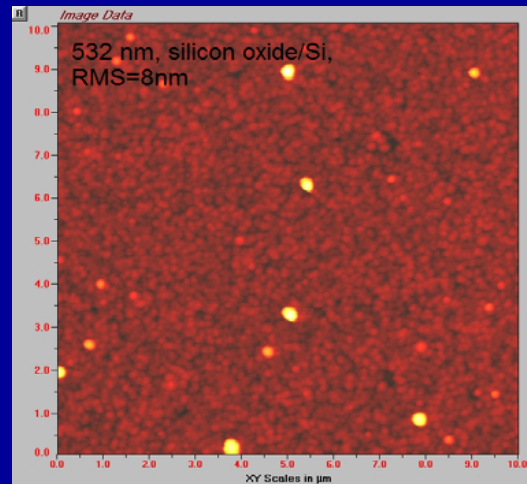
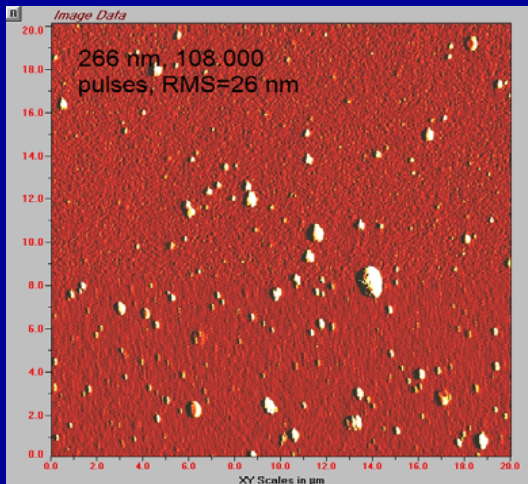
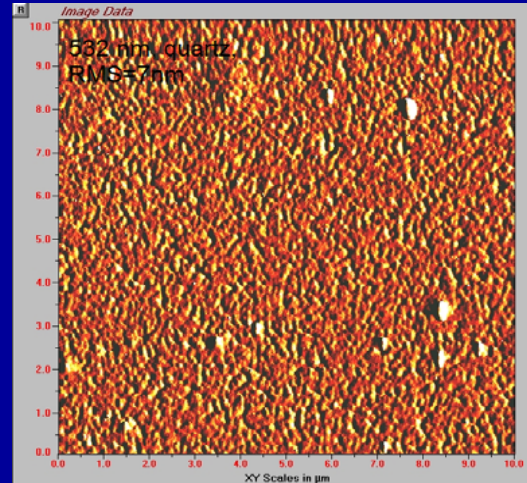
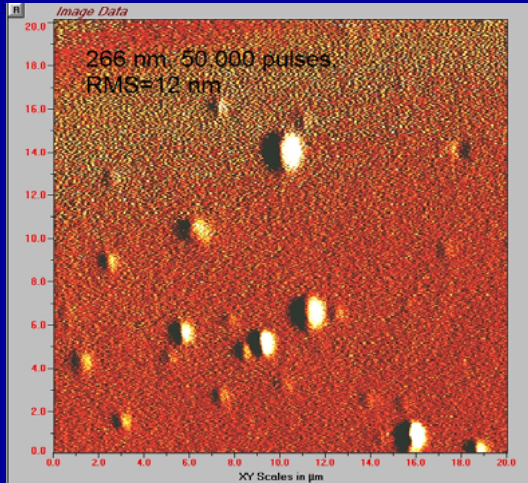
Influence of wavelength and substrate



→ The nature of the substrate is not significant.

→ The deposition temperature differs strongly according to the wavelength (150°C at 266 nm (UV) and 240°C at 532 nm (visible)).

Topography of n-type skutterudite films (AFM)



Quartz substrate:
influence of wavelength

→ The films exhibit a low amount of droplets, especially when made from 532 nm, and are smooth (RMS ~ 10 nm).

→ The surface shows a well defined morphology.

→ Grain sizes are about 100 to 200 nm.

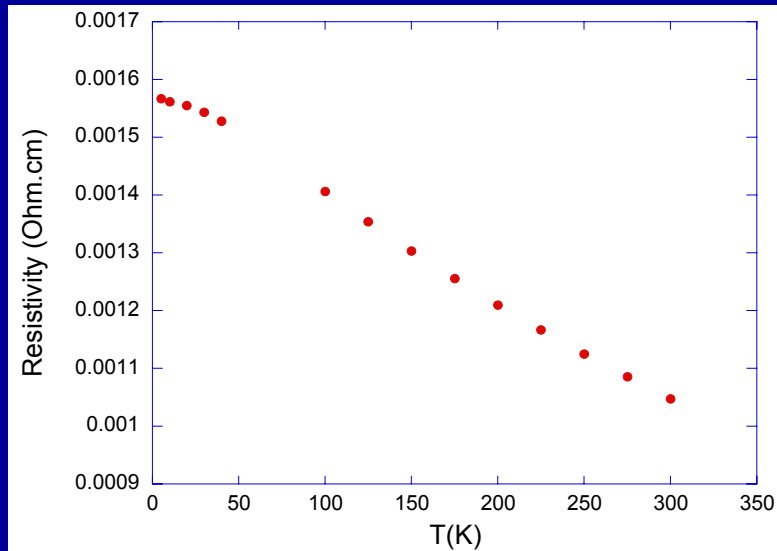
→ RMS does not depend on substrate nature.

Quartz substrate:
influence of thickness

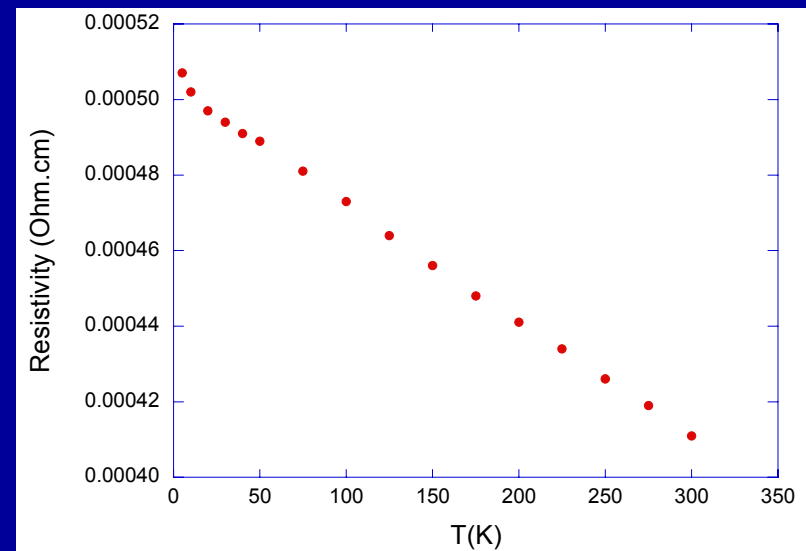
50 000 pulses:
influence of substrate

Electrical resistivity of skutterudite films

n-type



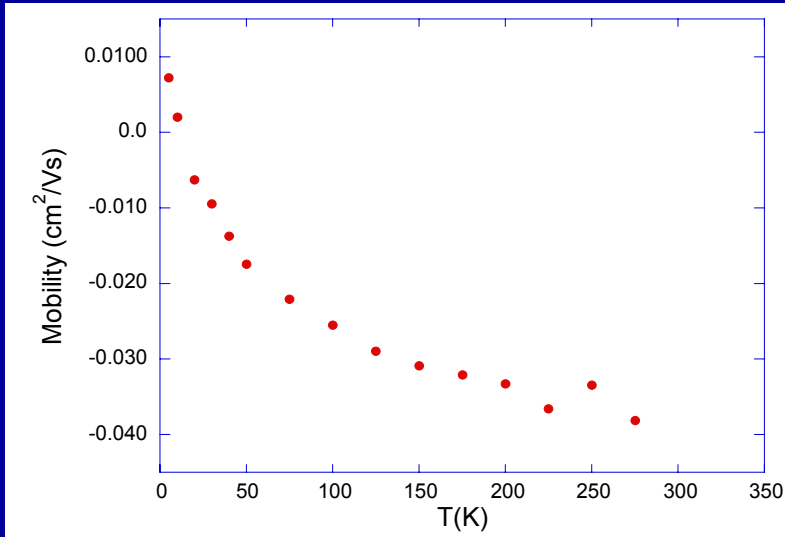
p-type



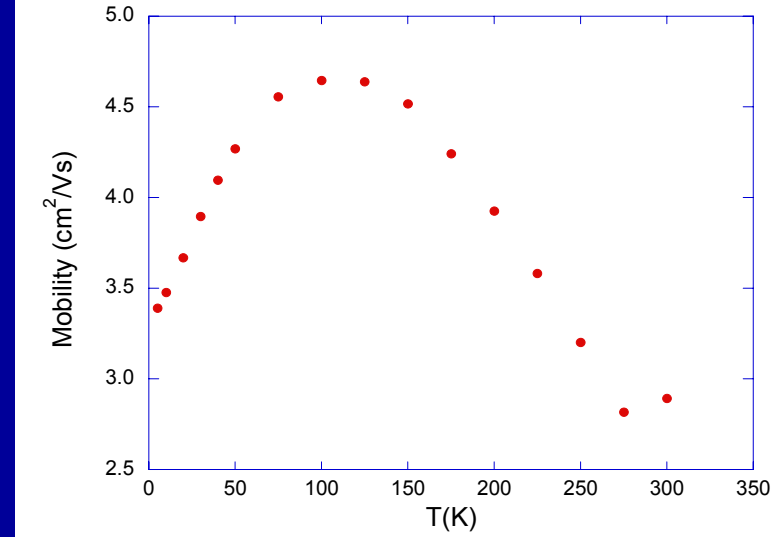
- Both *n*- and *p*-type materials show typical behaviours of semi-conductors.
- The values of the *p*-type film are much lower than those of the *n*-type film, in agreement with the differences observed for the bulk materials.

Carrier mobility of skutterudite films

n-type



p-type



→ The films exhibit the same type of conductivity as the target materials they are made from.

→ The preliminary results show that the carrier mobilities are as high as in bulk materials for the *p*-type films and much smaller than in bulk materials for the *n*-type films.

Skutterudite films: conclusions

→ The synthesis of skutterudite films revealed to be particularly sensitive to quite all deposition parameters we tested, but feasibility to make both *n*- and *p*-type materials by PLD has been proven.

→ The skutterudite phase could be achieved for the first time with the 532 nm wavelength, for a given density of energy (4 J/cm²), deposition temperature (240°C), and base pressure (10⁻⁴ mbar). These films exhibit less droplets and a smoother surface than films prepared in the UV range (for equivalent film thickness), contrarily to many materials.

→ The first transport property measurements showed that the films behaves similarly with temperature than the bulk materials.

Further work:

→ Try to realise a thermoelectric micro-generator made from both the *n* and *p*-type skutterudites synthesized.

Nanostructured bulk materials : Aim

Nano-structured bulk materials with enhanced thermoelectric performance

High yield production of nano-particles of thermoelectric materials

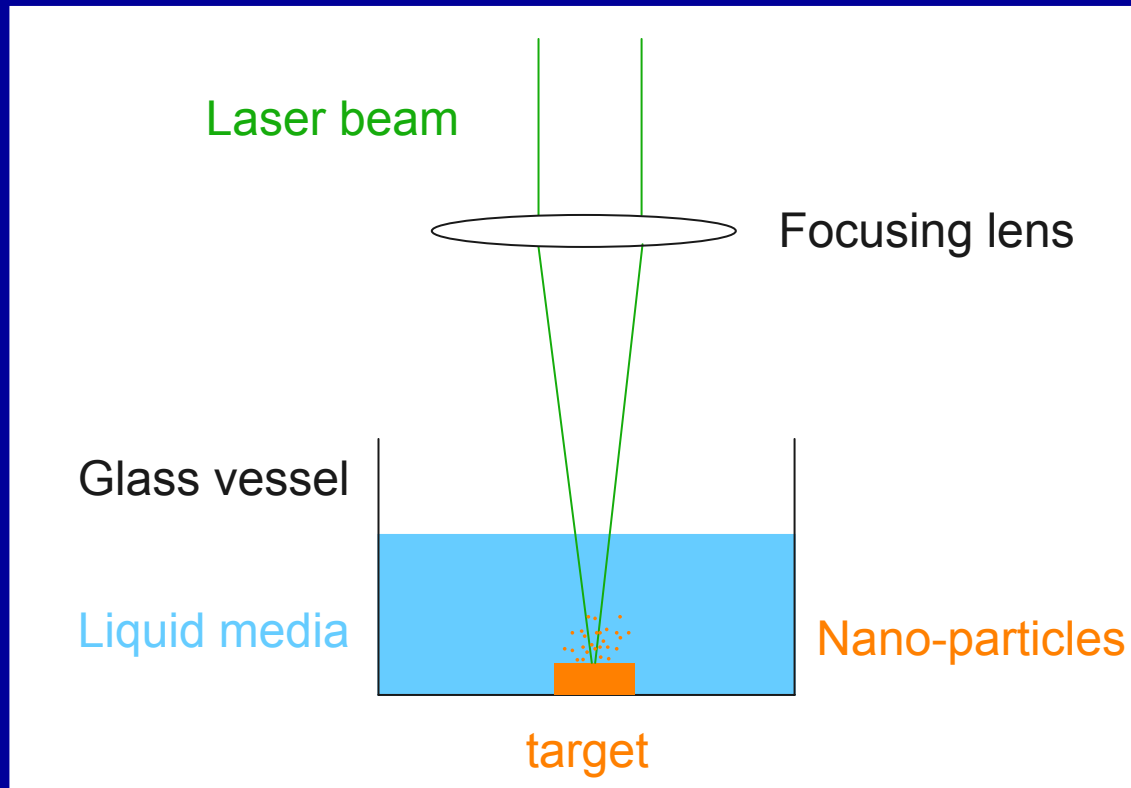
- size as small as possible
- narrow particle size distribution
- composition close to that of the starting material

Pulsed laser ablation in a liquid media
(simple, versatile, no chemical reagents)

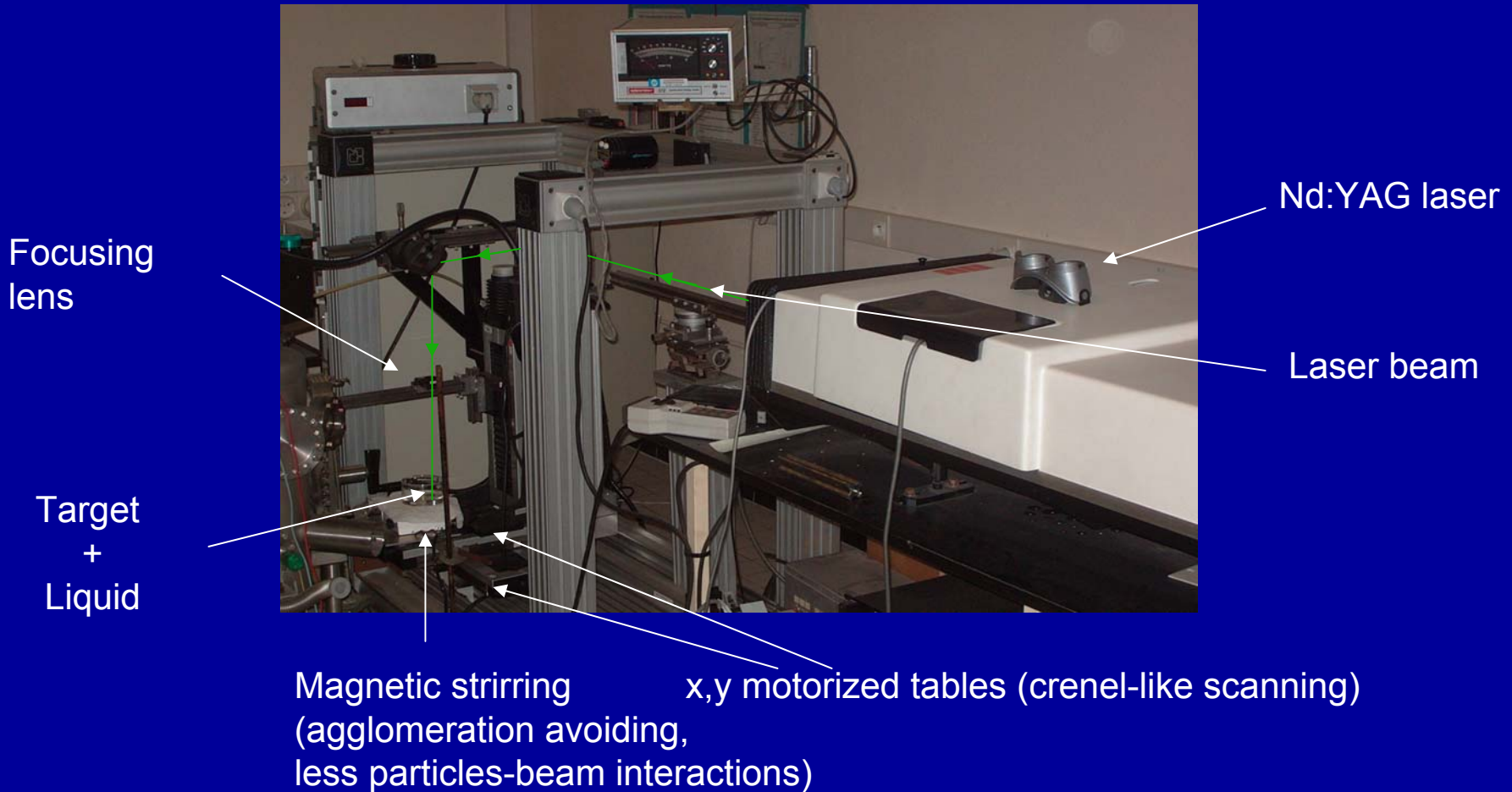
Materials: *n*- and *p*-type $(\text{Bi}_{1-x}\text{Sb}_x)_2(\text{Te}_{1-y}\text{Se}_y)_3$

Physico-chemical characterization of the produced powders

Pulsed laser ablation in a liquid media: principle



Experimental set-up

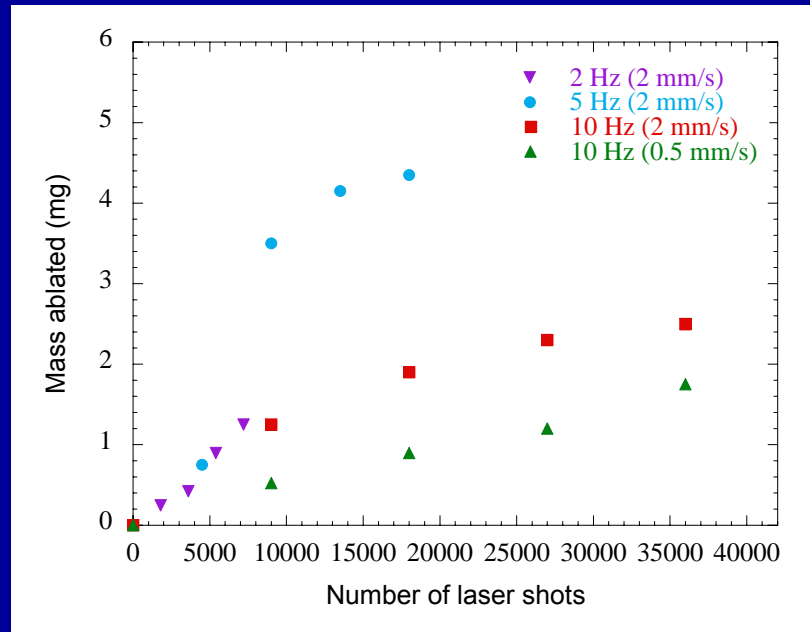


Experimental parameters

laser	532 or 1064 nm 2, 5 or 10 Hz 1 - 20 J/cm ² 1 - 36 000 shots
target	polycrystalline n (Bi _{0.95} Sb _{0.05}) ₂ (Te _{0.95} Se _{0.05}) ₃ p (Bi _{0.2} Sb _{0.8}) ₂ Te ₃
liquid	water , ethanol, n-heptane 1 or 2 cm height
scanning rate	0.5 or 2 mm/s

Yield optimization: influence of the laser frequency

n-type, 532 nm, water, 2.6 J/cm², 1 hour



Strong influence of the laser frequency

2 Hz: no overlapping between 2 consecutive shots

5 Hz: 50 % overlapping

10 Hz: 80 % overlapping (2 mm/s)

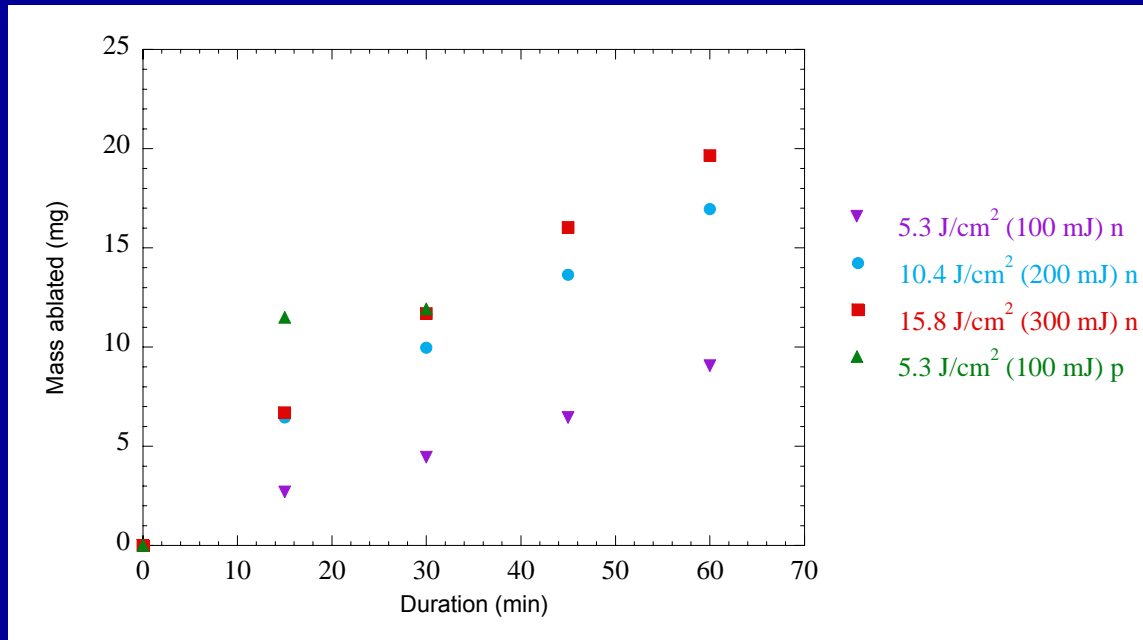
10 Hz: 90 % overlapping (0.5 mm/s)

→ 5 Hz, 2 mm/s

Saturation limit: ~ 3 mg → 70 μg/cm³

Yield optimization: influence of the fluence

1064 nm, water, 5 Hz



Density of energy \uparrow mass ablated \uparrow (limitation 300 mJ)

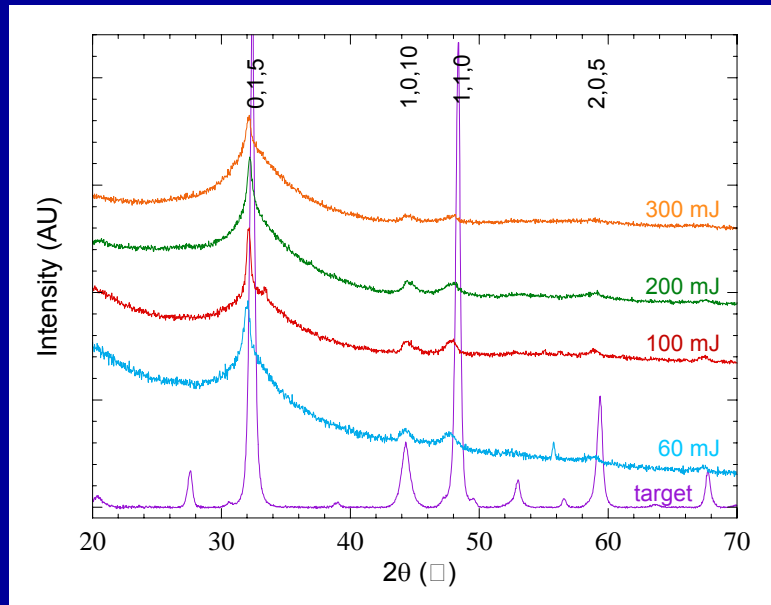
Saturation limit: ~ 10 mg $\rightarrow 220$ $\mu\text{g}/\text{cm}^3$ \rightarrow **IR**

p-type \neq *n*-type

Cristallographic structure: influence of the fluence

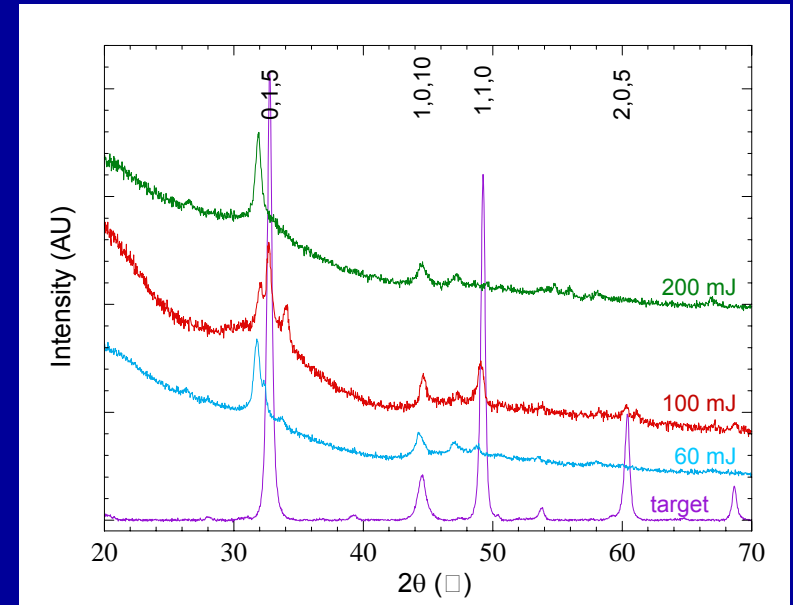
1064 nm, 18000 shots, 2 mm/s

***n*-type**



Achievement of the same phase as the target

***p*-type**



Achievement of a single phase but different from the target

or

Presence of multiple phases

Chemical composition (EPMA)

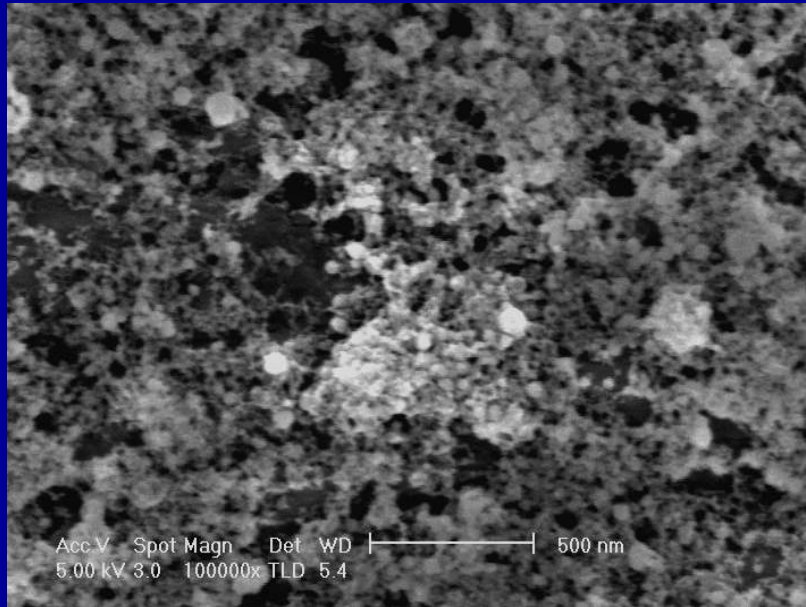
1064 nm, 18000 shots, 2 mm/s

	Bi	Te	Sb	Se
<i>n</i> -target	36.5	59.4	2.2	1.9
<i>n</i> -powders water, 100 mJ	38.2	58.2	1.8	1.8
<i>n</i> -powders water, 300 mJ	38.7	57.8	1.6	1.9
<i>p</i> -target	7.6	60.7	31.7	-
<i>p</i> -powders water, 60 mJ	8.0	60.8	31.3	-

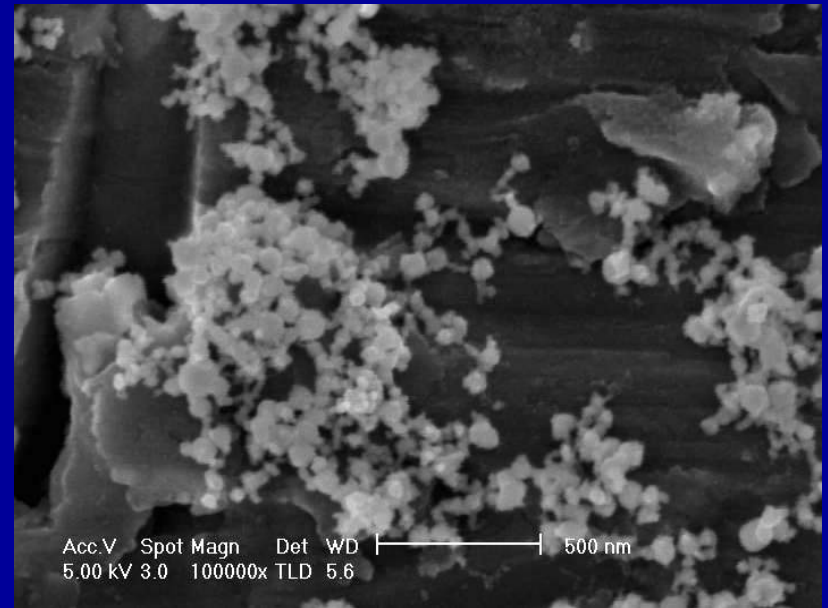
Morphology

1064 nm, 18000 shots, 60 mJ, 2 mm/s

n-type



p-type



Mean particle size

TEM: 28 nm (200 shots)

XRD: 35 nm (18000 shots)

Summary

Proof-of-principle: pulsed laser ablation in a liquid media is efficient to synthesize nano-powders of complex materials.

n-type $(\text{Bi}_{0.95}\text{Sb}_{0.05})_2(\text{Te}_{0.95}\text{Se}_{0.05})_3$ can be synthesized in water.

p-type $(\text{Bi}_{0.2}\text{Sb}_{0.8})_2\text{Te}_3$ is more difficult to synthesize.

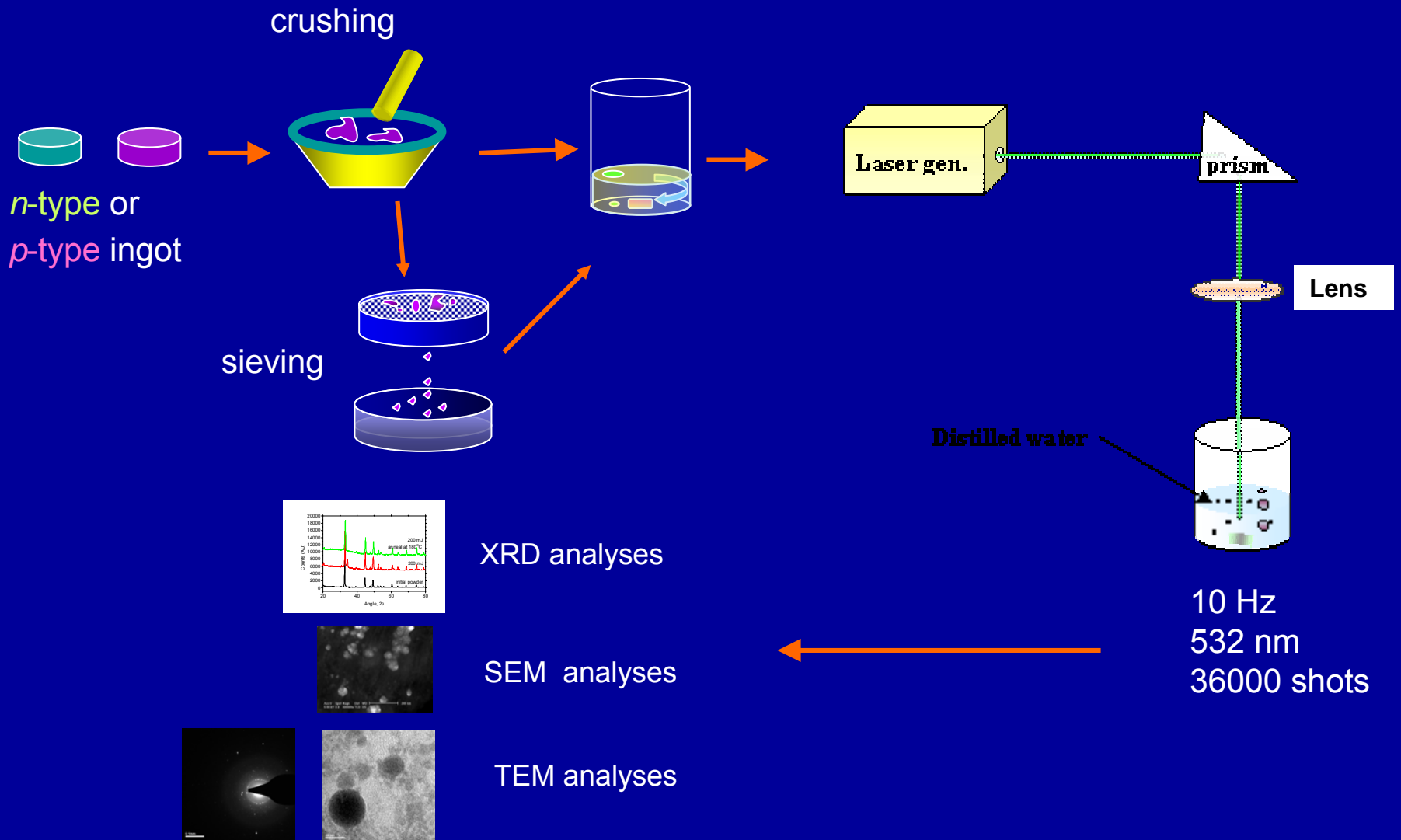
(different absorption coefficient, different interaction with the solvent).

Each parameter studied has an influence (tailor as a function of we want):

- wavelength → saturation limit of particles in the solution, size, composition
- solvent → height: yield,
nature: crystallographic phase, size, agglomeration aptitude
- energy → yield, size, crystallographic phase

Problems: low ablation yield, no *p*-type, 'large' particle size distribution (laser-powder interaction ?), inflammability and recuperation of the solvent, oxidation.

New process diagram

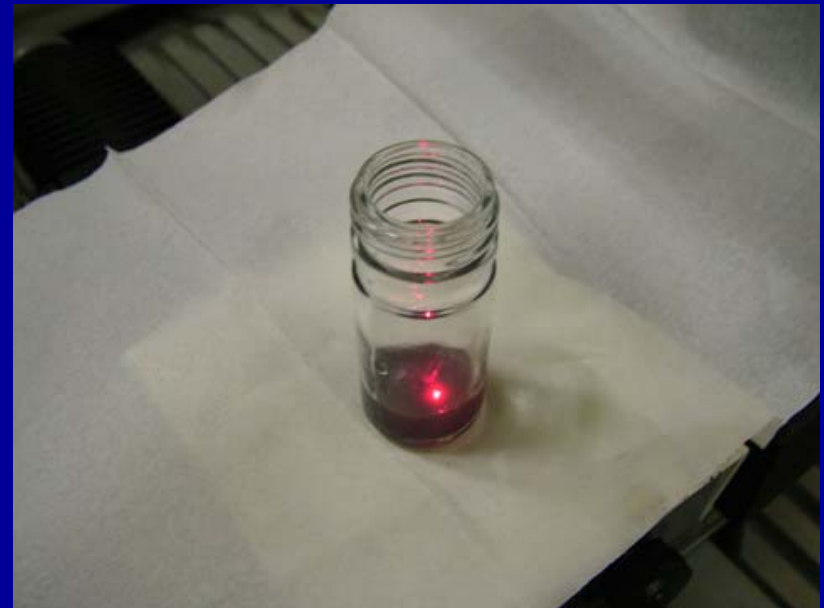


New experimental



Initial powders in distilled water
(small diameter vessel, 10 Hz)

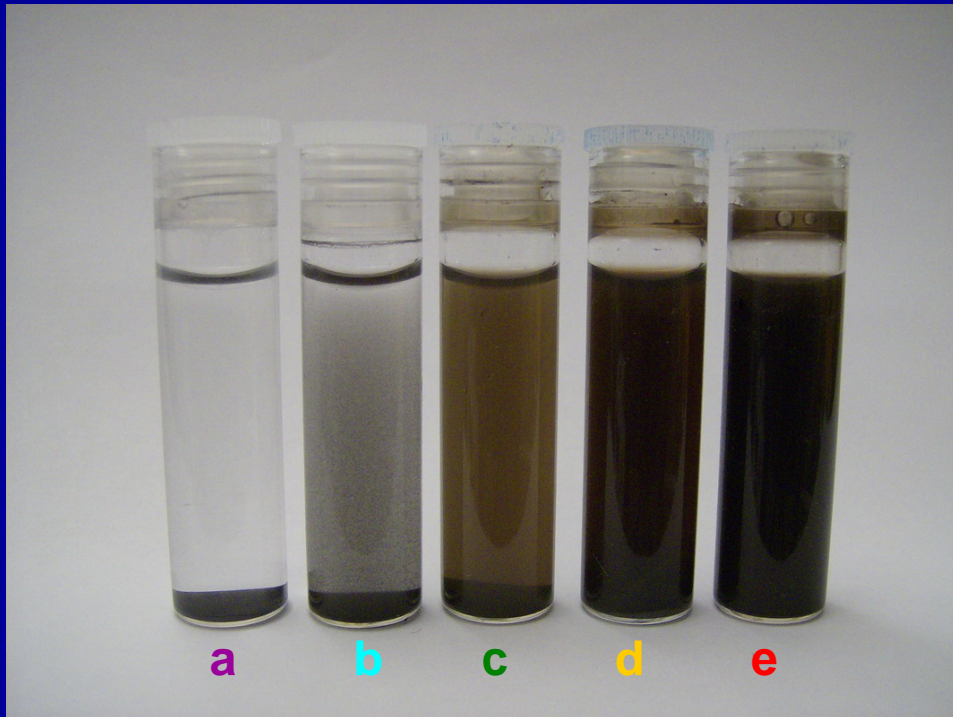
→ 3 magnets



Guiding beam for adjusting the laser
beam position (excentered position)

Influence of the number of shots: sedimentation test

p-type, 532 nm, 300 mJ, 30 min test



a: initial powders

b: 9 000 shots

c: 18 000 shots

d: 27 000 shots

e: 36 000 shots

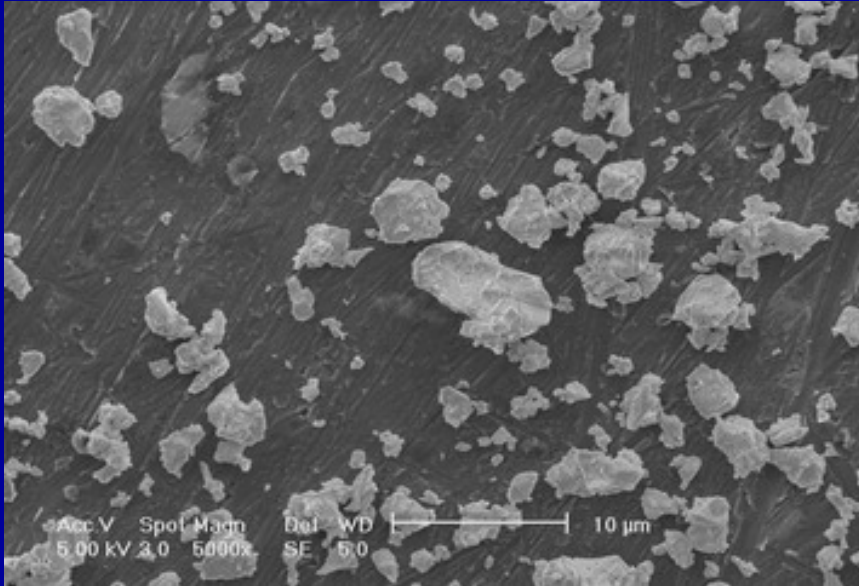
Duration of sedimentation increases as a function of the number of shots:

→ the weight and therefore the size of the generated particles become lower,

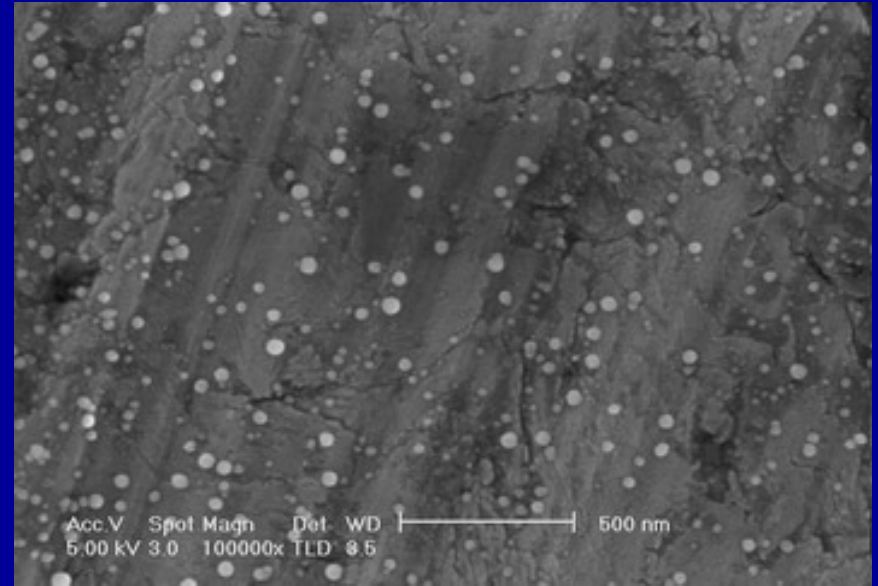
→ more and more initial particles are broken.

Effect of the laser beam on the particles morphology

p-type, 532 nm, 300 mJ, 36 000 shots



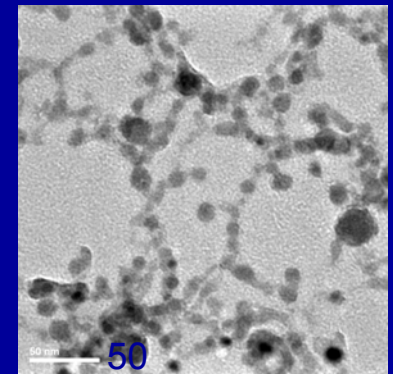
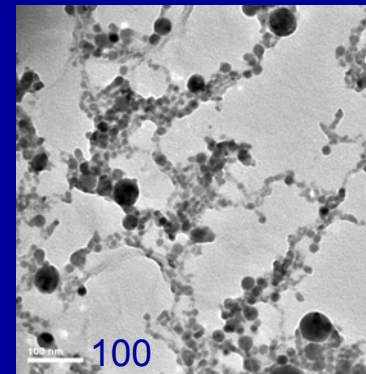
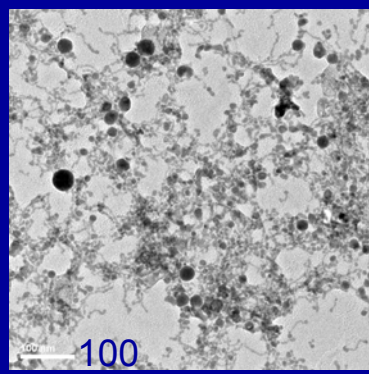
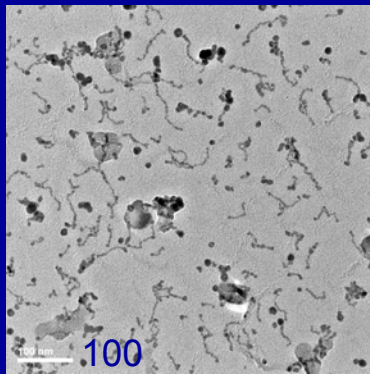
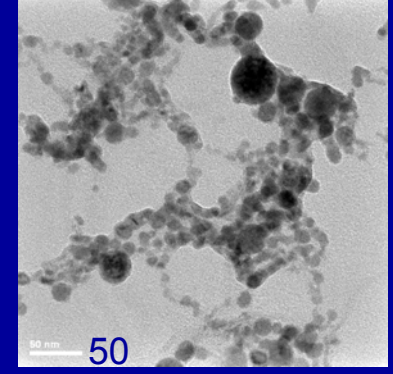
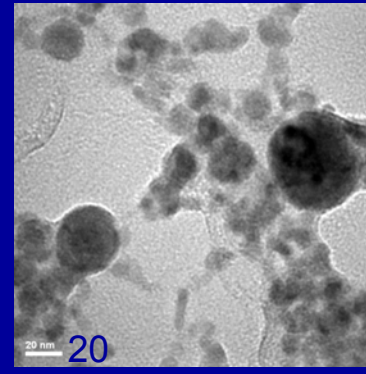
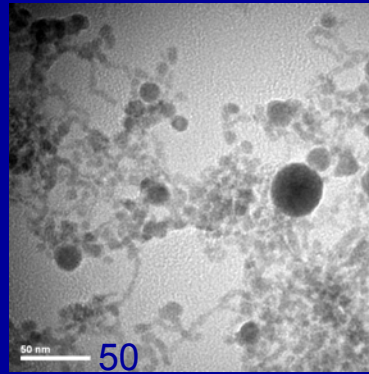
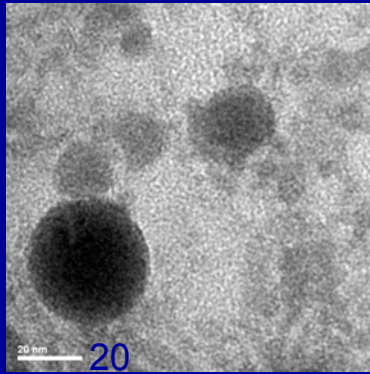
Initial sieved powders: diameter is in the range of 1- 17 μm and 2.5 μm in average



After laser treatment: nano-powders of size less than 30 nm

Influence of the composition on the particle size

532 nm, 200 mJ, 36 000 shots



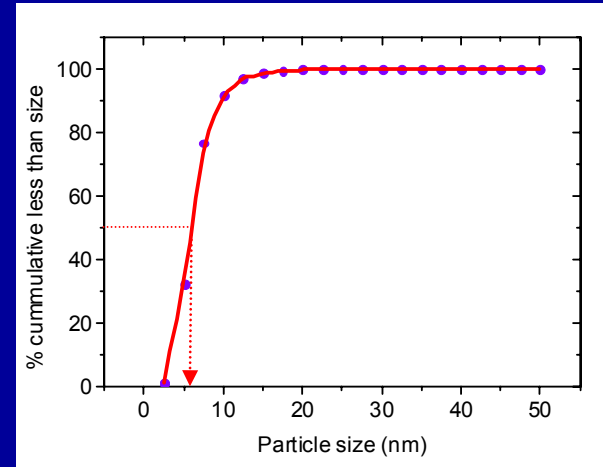
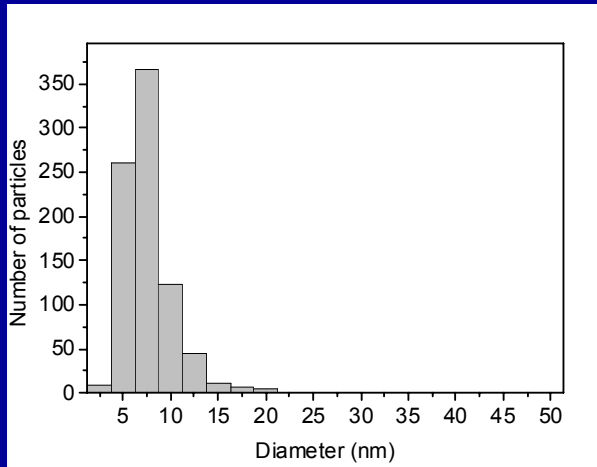
n-type nano-powders

p-type nano-powders

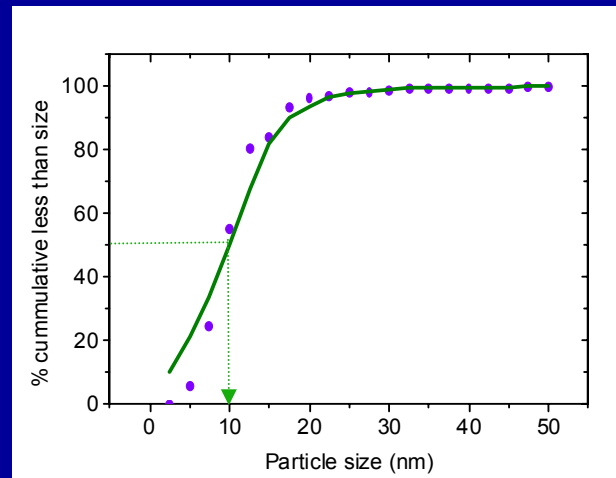
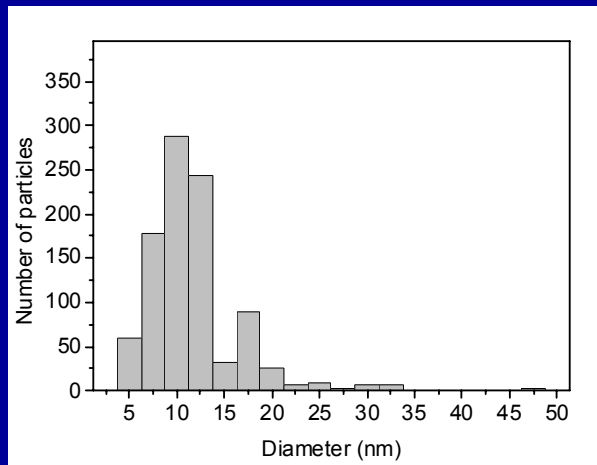
Influence of the composition on the particle size

532 nm, 200 mJ, 36 000 shots

n-type: diameter in the range of 2.5- 22.5 nm and 6 nm in average



p-type: diameter in the range of 2.5 – 47.5 nm and 10 nm in average

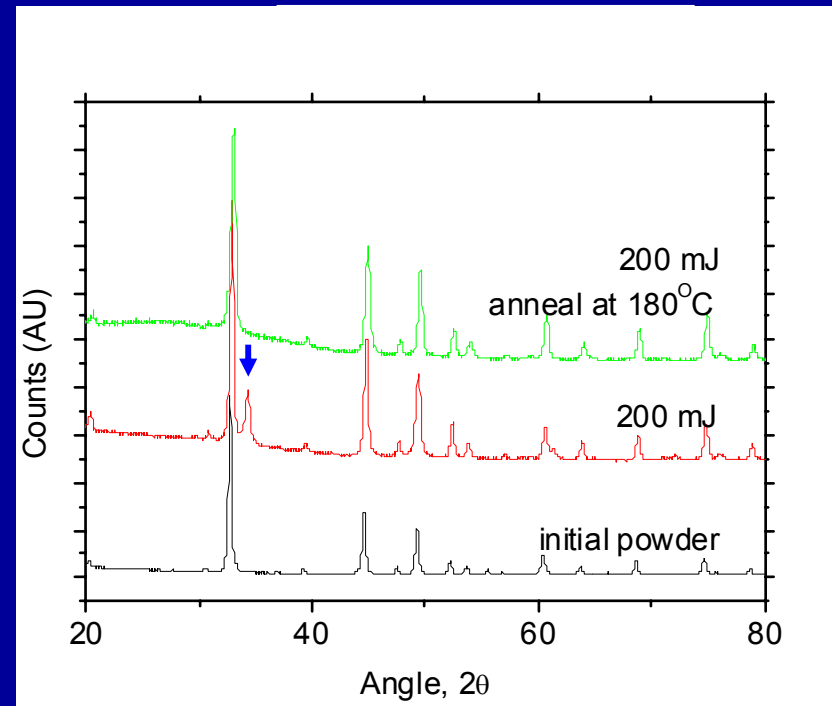
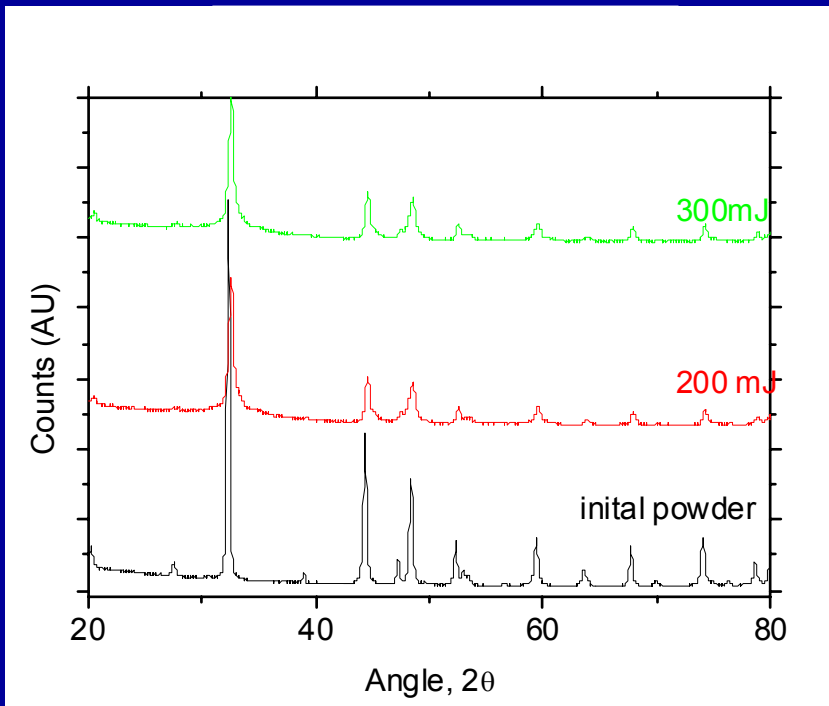


Crystallographic structure

532 nm, 36 000 shots

n-type $(\text{Bi}_{0.95}\text{Sb}_{0.05})_2(\text{Te}_{0.95}\text{Se}_{0.05})_3$

p-type $(\text{Bi}_{0.2}\text{Sb}_{0.8})_2\text{Te}_3$



- no significant difference as a function of output energy for both type
- *n*-type: single phase
- *p*-type: unknown phase, disappearing after annealing at 180°C

Conclusion

By comparison to the production of nano-powders from a bulk target, the use of initial micro-sized powders leads to:

→ smaller particles,

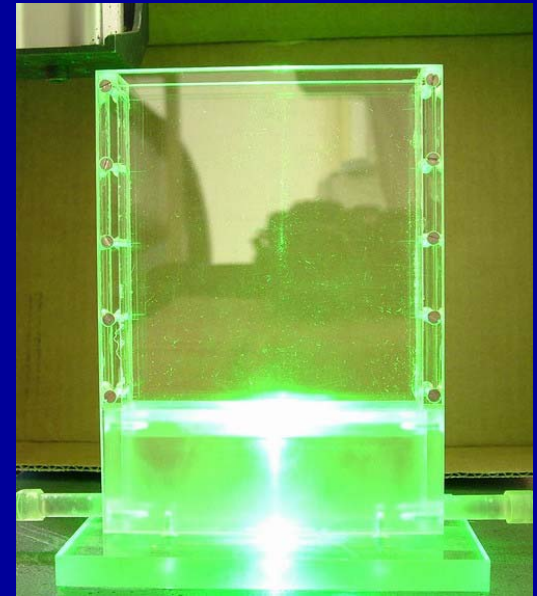
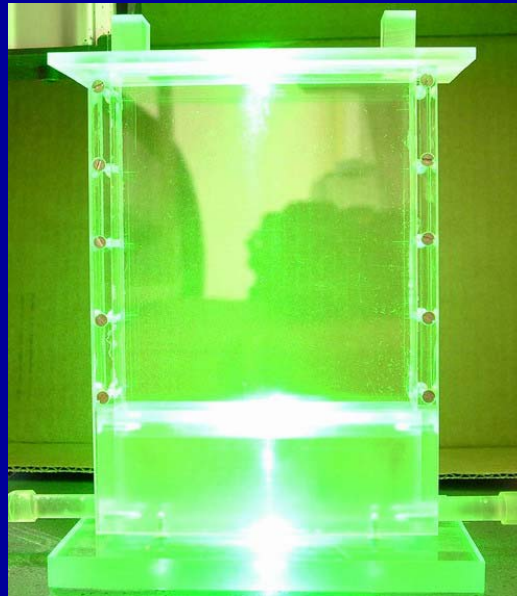
→ improved production yield,

→ improved crystalline quality of the p-type nano-powders (annealing),

→ no inflammability problem (use of water).

Now

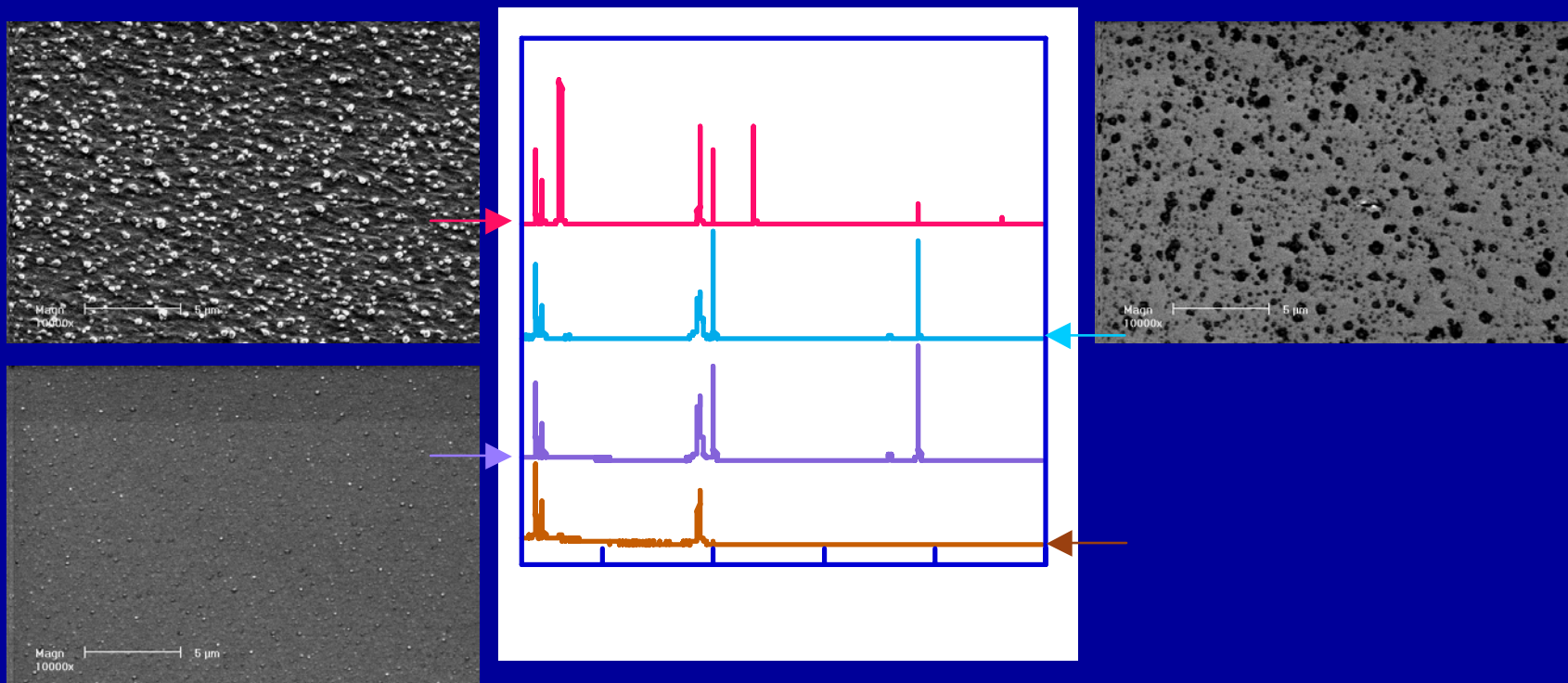
Acrylic box with powder circulation in water



Use of a new fabrication cell to produce nano-powders with high yield to make:

- nano-structured bulk materials (SPS) to test the thermoelectric performance (electrical and thermal conductivities, thermopower, →improvement?)
- thin films directly from the solution by electrophoresis and test their thermoelectric performance (use in micro-devices: μ -generators or μ -refrigerators)

PbTe-Bi nano-composites: *influence of bi-layers number (BaF₂, 150°C)*



- Smoothing of the surface as the number of bi-layers increases
- Obtaining of the (111) texture

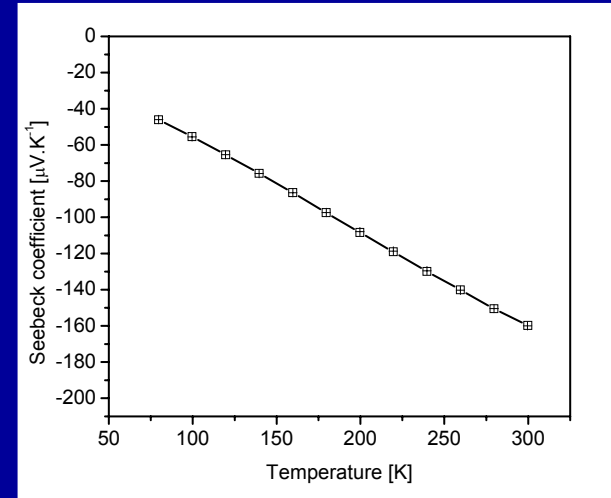
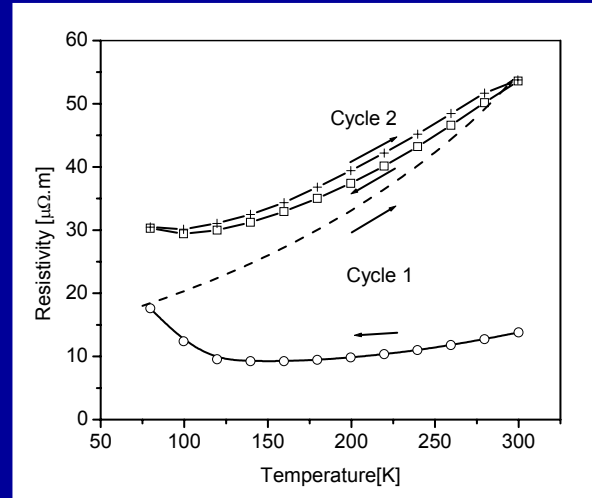
Transport properties of PbTe films and Bi/PbTe nano-composites

($T_s = 150^\circ\text{C}$, $F = 4 \text{ J/cm}^2$)

	Resistivity [$\mu\Omega\cdot\text{m}$]	Seebeck [$\mu\text{V}\cdot\text{K}^{-1}$]	Power factor [$\mu\text{W}\cdot\text{m}^{-1}\cdot\text{K}^{-2}$]
PbTe/BaF ₂	45	- 247	1355 ($n=4.9\times 10^{17} \text{ cm}^{-3}$)
20 (PbTe/Bi/BaF ₂)	32	- 223	1554 ($n=2.3\times 10^{18} \text{ cm}^{-3}$)
PbTe/glass	14	- 156	1760 ($n=2.0\times 10^{20} \text{ cm}^{-3}$)
20(PbTe/Bi/glass)	8	- 118	1780

Thermal cycling of PbTe films and Bi/PbTe nanocomposites

The PbTe films do not withstand thermal cycling ...



...but the Bi/PbTe nanocomposites do !

